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Order reconstruction patterns in nematic liquid crystal wells

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Abstract

We numerically study structural transitions inside shallow sub-micron scale wells with square cross-section, filled with nematic liquid crystal material. We model the wells within the Landau-de Gennes theory. We obtain two qualitatively different states: (i) the DSD (diagonal state with defects) for relatively large wells with lateral dimension greater than a critical threshold and (ii) a new, two-dimensional star-like biaxial order reconstruction pattern called the WORS (well order reconstruction structure), for wells smaller than the critical threshold. The WORS is defined by an uniaxial cross connecting the four vertices of the square cross-section. We numerically compute the critical threshold in terms of the bare biaxial correlation length and study its dependence on the temperature and on the anchoring strength on the lateral well surfaces.

1 Introduction

Liquid crystals (LCs) are mesogenic phases of matter with physical properties intermediate between those of the conventional solid and liquid phases of matter [2]. Nematic LCs are the simplest type of LCs; nematics are complex anisotropic liquids with a certain degree of long-range orientational ordering. The structural properties of a nematic LC can be easily manipulated by incident light, electric or magnetic fields and mechanical effects [14]. As a consequence, they exhibit a wealth of diverse physical phenomena and are often used as a testing ground for fundamental physics in condensed matter systems [7]. Nematic LCs now have widespread applications across science and technology e.g. electro-optic devices, sensors and notably, the huge and thriving liquid crystal display (LCD) industry around the globe [20, 23, 24].

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Nematics in confinement have generated tremendous interest in recent years [11, 14, 30, 28]. Such confined LC systems typically exhibit topological defects or localized material imperfections [22, 24]. Defect core structures are poorly understood; experimental and numerical studies indicate that defect cores can be associated with uniaxial-biaxial structural transitions i.e. structural transitions from a phase with a single distinguished direction (uniaxial) to a phase with two distinguished directions (biaxial) [14, 28]. Such structural transitions were first reported in the benchmark paper [28]. In [28], the authors numerically analyze the interior of a defect core; the nematic LC is in a uniaxial phase at the defect core and in a uniaxial phase away from the defect core. The corresponding directors/distinguished directions (see (2)) are perpendicular to one another and the LC material mediates between the two conflicting uniaxial states via a continuum of biaxial states. These transitions, often referred to as "order reconstruction" (OR) phenomena were subsequently reported in a batch of numerical papers [6, 3, 16, 17, 31] which indicate that OR phenomena typically occur in severely confined systems with comparable material and geometrical length scales or/and under the action of intense electric fields. For example, in [25], the authors study the OR phenomenon for a twisted hybrid nematic cell. The authors numerically find two classes of solutions: (i) the conventional purely uniaxial twisted solution wherein the molecules twist uniformly between the two boundary plates and (ii) the OR untwisted solution characterized by a biaxial band around the centre of the cell. The OR solution is the only observable solution in nano-confined geometries where the cell gap is comparable to a material-dependent length scale, known as the biaxial correlation length [6]. Experimental studies of the OR phenomenon have been reported in for e.g. [3, 12] and such studies are particularly relevant for new LC systems with intrinsic biaxiality and their potential applications in nano-science and technology.

Our work is motivated in part by the results in [20, 29] with a view to characterize new two-dimensional (2D) OR patterns in prototype geometries. This is an interesting problem since classical OR patterns are typically one-dimensional (1D) with the structural characteristics varying along one coordinate direction, e.g. along the radial direction in a spherical droplet (see [28]), along the vertical direction in a cylinder [16, 17] or along the normal direction to parallel bounding plates [25] etc. and 2D OR patterns offer new scientific possibilities. In [20, 29], the authors study a model liquid crystal device comprising a periodic array of shallow micron-sized square wells. These square wells are filled with nematic LC; the surfaces are treated so as to induce tangential or planar boundary conditions [20, 23] i.e. on the well surfaces, the LC molecules are constrained to be in the plane of the surfaces. This relatively simple geometry is bistable or multistable in the sense that there are multiple stable LC textures with contrasting optical properties [29]. In [20, 29], the authors work within the Landau-de Gennes theory for

nematic LCs [7]. Given that the wells are shallow, they argue that it suffices to model 2D structural variations within the plane of the square and work with a strictly 2D model i.e. study 2D configurations on the square. They work with micron-size wells so that the square size, denoted by R , is much greater than a material length-scale known as the bare biaxial correlation length [16, 17]. The bare biaxial correlation length is a measure of the correlation in nematic order, at the nematic-isotropic transition temperature, and is typically a few tens of nanometers (see [3, 10, 30]). In this limit, the stable LC states are effectively uniaxial almost everywhere, except for small localized defects around the square vertices [20]. The authors find two stable LC states: (i) the diagonal state with the average uniaxial alignment along one of the square diagonals and (ii) the rotated state where the average direction of alignment rotates by 180 degrees between a pair of opposite square edges.

We carefully examine the effects of severe confinement on the same model system. In particular, we work with a three-dimensional (3D) model, under the assumption that all structural details are independent of the vertical coordinate. Our 3D modelling incorporates biaxiality, which is outside the scope of the 2D model employed in [29, 20]. We measure the square cross-section size, R , in units of the bare biaxial correlation length, $\xi_b^{(0)}$, and obtain two qualitatively different LC textures as a function of R : (i) DSD for large R and (ii) the WORS for small R . The DSD is qualitatively similar to the diagonal solution in [29, 20] and our 3D modelling illustrates the biaxial structure of the defect cores around the square vertices. The WORS is a new 2D OR pattern connecting the four square vertices characterized by a star-shaped rim of maximal biaxiality connecting the four square vertices. The biaxial rim separates a negatively ordered uniaxial state along the square diagonal from the positively ordered uniaxial states near the square edges. In other words, the system mediates between different uniaxial states (the positively ordered boundary states and the negatively ordered diagonal state) via a 2D star-shaped ring of intermediate biaxial states. The WORS pattern is the only stable LC pattern for small wells, with $R \leq R_c$, where R_c is a critical threshold length. In Section 4, we compute R_c in units of $\xi_b^{(0)}$, systematically investigate how R_c depends on temperature and anchoring strength on the lateral boundaries, to obtain quantitative information about the stability of the WORS.

The paper is organized as follows. In Section II, we review the Landau-de Gennes theory for nematic LCs. In Section III, we outline the model geometry, the relevant parametrizations, the material length-scales and the governing Euler-Lagrange equations. In Section IV, we present our numerical results and summarize our conclusions and future outlook in Section V.

2 Theoretical background

Let $\mathcal{B} \subset \mathbb{R}^3$ denote an arbitrary three-dimensional domain filled with nematic LC material, with a sufficiently regular boundary, $\partial\mathcal{B}$. We work within the Landau-de Gennes (LdG) theory for nematic LCs wherein the state of the nematic LC is described by a macroscopic order parameter, the \mathbf{Q} -tensor parameter [2, 7]. The \mathbf{Q} -tensor order parameter can be viewed as a macroscopic measure of the system anisotropy i.e. a measure of the anisotropy in dielectric response to electric fields or anisotropy in magnetic susceptibility or response to magnetic fields. Mathematically, \mathbf{Q} is a symmetric, traceless 3×3 matrix and from the spectral decomposition theorem, the \mathbf{Q} -tensor can be written in terms of its eigenvalues $\{s_i\}$, and eigenvectors, $\{\vec{e}_i\}$, as shown below [14]

$$\mathbf{Q} = \sum_{i=1}^3 s_i \vec{e}_i \otimes \vec{e}_i. \quad (1)$$

The eigenvectors represent the distinguished directions of LC alignment and the eigenvalues are a measure of the degree of orientational ordering about these directions. A LC phase is said to be in the (i) disordered or isotropic phase if $\mathbf{Q} = 0$, (ii) uniaxial phase if \mathbf{Q} has a pair of degenerate non-zero eigenvalues and (iii) the biaxial phase if \mathbf{Q} has three distinct eigenvalues [7]. In the uniaxial case, \mathbf{Q} can be concisely expressed as

$$\mathbf{Q} = S \left(\vec{e}_1 \otimes \vec{e}_1 - \frac{1}{3} \mathbf{I} \right), \quad (2)$$

in terms of the single distinguished eigendirection, the *director field* \vec{e}_1 , and the scalar order parameter S . In other words, \vec{e}_1 , is the eigenvector with the non-degenerate eigenvalue and all directions perpendicular to \vec{e}_1 are physically equivalent. In the biaxial phase, \mathbf{Q} can be written as

$$\mathbf{Q} = S \left(\vec{e}_1 \otimes \vec{e}_1 - \frac{1}{3} \mathbf{I} \right) + U (\vec{e}_2 \otimes \vec{e}_2 - \vec{e}_3 \otimes \vec{e}_3), \quad (3)$$

where $\{\vec{e}_i\}$ are the eigenvectors and (S, U) are two scalar order parameters that measure the degree of orientational ordering about the different eigenvectors [21]. A quantitative measure of the degree of biaxiality is defined to be

$$\beta^2 = 1 - \frac{6(\text{tr} \mathbf{Q}^3)^2}{(\text{tr} \mathbf{Q}^2)^3}, \quad (4)$$

$\beta^2 \in [0, 1]$ and $\beta^2 = 0$ for all uniaxial states [23]. The maximal value, $\beta^2 = 1$, is attained if and only if one of the eigenvalues, s_i , vanishes.

The LdG theory is a variational theory and the stable physically observable LC states correspond to minimizers of an appropriately defined LdG

energy functional [21, 23]. We work with the following LdG free energy

$$F = \int_{\mathcal{B}} (f_c + f_e) d^3\vec{r} + \sum_i \int_{\partial\mathcal{B}} f_s^{(i)} d^2\vec{r}, \quad (5)$$

where $d^3\vec{r}$ and $d^2\vec{r}$ denote the volume and area measures, respectively [14, 19]. The condensation (f_c), elastic (f_e), and surface ($f_s^{(i)}$) free energy densities are given by [14]

$$f_c = \frac{A_0(T - T_*)}{2} \text{tr} \mathbf{Q}^2 - \frac{B}{3} \text{tr} \mathbf{Q}^3 + \frac{C}{4} (\text{tr} \mathbf{Q}^2)^2, \quad (6)$$

$$f_e = \frac{L}{2} |\nabla \mathbf{Q}|^2, \quad (7)$$

$$f_s^{(i)} = \frac{w^{(i)}}{2} \text{Tr} \left(\mathbf{Q} - \mathbf{Q}_s^{(i)} \right)^2. \quad (8)$$

In particular, we only focus on static equilibria and do not include external fields in this variational problem. The condensation energy dictates the preferred degree of orientational order in the bulk as a function of the temperature. In (6), A_0 , B , and C are material constants, T denotes the temperature and T_* is a characteristic material-dependent supercooling temperature. One can readily check that all stationary points of f_c are necessarily uniaxial and that there are three characteristic temperatures: (i) $T = T_*$ below which the isotropic phase is unstable, (ii) $T_{IN} = T_* + \frac{B^2}{27A_0C}$ associated with a first-order isotropic-nematic phase transition and (iii) the nematic superheating temperature, $T_{**} = T_* + \frac{B^2}{24A_0C}$, above which ordered nematic states do not exist. For temperatures $T < T_{IN}$, the condensation enforces a uniaxial scalar order parameter, $S = S_{eq}(T) = \frac{B + \sqrt{B^2 - 24AC}}{4C}$, in the bulk and for temperatures, $T > T_{IN}$, f_c pushes the system towards the disordered isotropic phase. The elastic energy density penalizes any spatial inhomogeneities in the system and we work within the simplest one-constant approximation in (7). The elastic constant is denoted by $L > 0$ in (7). The surface anchoring energy density, $f_s^{(i)}$, enforces a preferred orientation, encoded by $\mathbf{Q}_s^{(i)}$, on the i -th lateral surface. The anchoring coefficient, $w^{(i)}$, is a measure of the strength of the surface anchoring and the $w^{(i)} \rightarrow \infty$ limit is the strong anchoring limit (also see [20] for comparisons between strong and weak anchoring). Whilst the elastic energy coerces the system towards a spatially homogeneous or uniform state, the condensation energy promotes uniform uniaxial bulk order given by $S = S_{eq}(T)$. However, the boundary conditions typically require that both conditions cannot be simultaneously satisfied in confined geometries, resulting in inhomogeneous pattern formation. In what follows, we compare the competing effects of the elastic contributions, condensation term and the surface effects by means of two different length scales: a material-dependent length scale known as the bare biaxial correlation length and the surface extrapolation length [16, 3].

3 Modelling framework

3.1 Geometry

We study nematic samples inside a 3D well with square cross-section, $\mathcal{B} \subset \mathbb{R}^3$, given by

$$\mathcal{B} = \{(x, y, z) \in \mathbb{R}^3; 0 \leq x, y \leq R, 0 \leq z \leq h\}. \quad (9)$$

In (9), h is the well height, R is the lateral dimension and $h \ll R$, consistent with the assumptions in [29, 20]. We take $h = \frac{R}{10}$.¹ The coordinate unit-vectors are denoted by $(\vec{e}_x, \vec{e}_y, \vec{e}_z)$. We assume that the structural characteristics only depend on the (x, y) -spatial coordinates and are independent of the vertical z -coordinate; an assumption often used for shallow wells [20, 29]. The boundary plates are located at $\{x = 0, R\}$, $\{y = 0, R\}$, $\{z = 0, h\}$ and all plates are treated to induce tangent or planar degenerate boundary conditions. The tangent boundary conditions are implemented using a combination of strong and weak anchoring and natural boundary conditions.² On the lateral surfaces, $x = \{0, R\}$ and $y = \{0, R\}$, the tangent conditions are implemented via Dirichlet conditions/strong anchoring or weak anchoring/surface energies. In the strong anchoring case, the corresponding \mathbf{Q} -tensor is prescribed to be strictly uniaxial, with director parallel to \vec{e}_x and \vec{e}_y , on the xz and yz -surfaces respectively. We impose natural boundary conditions on the top and bottom faces, $z = 0$ and $z = h$. All numerical simulations are initialized using planar initial conditions. The planar initial conditions, combined with the natural boundary conditions, bias the solutions to respect the planar degenerate conditions on the top and bottom plates so that no particular direction in the (x, y) -plane is distinguished or singled out.

3.2 Parametrization

We adopt the following 3D parametrization of the \mathbf{Q} -tensor order parameter, as given in [16, 5]

$$\mathbf{Q} = (q_3 + q_1)\vec{e}_x \otimes \vec{e}_x + (q_3 - q_1)\vec{e}_y \otimes \vec{e}_y + q_2(\vec{e}_x \otimes \vec{e}_y + \vec{e}_y \otimes \vec{e}_x) - 2q_3\vec{e}_z \otimes \vec{e}_z, \quad (10)$$

where q_1 , q_2 , and q_3 are independent of z and only depend on x and y . We assume $\vec{e}_3 = \vec{e}_z$ is always an eigenvector of \mathbf{Q} . The remaining two eigenvectors are allowed to rotate with respect to the reference frame, (\vec{e}_x, \vec{e}_y) ,

¹The trends would be unchanged for heights $\frac{R}{10} \leq h \leq \frac{R}{5}$.

²The tangent conditions may be violated for energetic considerations.

by an angle $\varphi \in [0, \pi)$. We have

$$\begin{aligned}\vec{e}_1 &= \cos \varphi \vec{e}_x + \sin \varphi \vec{e}_y, \\ \vec{e}_2 &= -\sin \varphi \vec{e}_x + \cos \varphi \vec{e}_y,\end{aligned}\tag{11}$$

$$\vec{e}_3 = \vec{e}_z.\tag{12}$$

The parameter q_2 is a measure of the departure of the eigenframe, $(\vec{e}_1, \vec{e}_2, \vec{e}_3)$, from the Cartesian frame, $(\vec{e}_x, \vec{e}_y, \vec{e}_z)$ and the two frames coincide when $q_2 = 0$. The eigenvalues $\{s_i\}$ are explicitly given by $s_1 = q_3 + \sqrt{q_1^2 + q_2^2}$, $s_2 = q_3 - \sqrt{q_1^2 + q_2^2}$ and $s_3 = -2q_3$. In particular, the biaxiality parameter is defined to be

$$\beta^2(q_1, q_2, q_3) = 1 - 6 \frac{\left(6\sqrt{(q_1^2 + q_2^2)}q_3 - 6q_3^3\right)^2}{(6q_3^2 + 2(q_1^2 + q_2^2))^3}.\tag{13}$$

The condition, $\sqrt{q_1^2 + q_2^2} = 0$, corresponds to the exchange of eigenvalues, $s_1 \leftrightarrow s_2$, in the nematic phase [28].

We briefly compare the parametrization (10) with the parametrization in [20] where the authors study strictly two-dimensional LC configurations on a square domain; the \mathbf{Q} -tensor is then given by

$$\mathbf{Q} = S(\vec{e}_1 \otimes \vec{e}_1 - \vec{e}_2 \otimes \vec{e}_2).\tag{14}$$

Here, $\text{tr} \mathbf{Q}^3 = 0$, the biaxiality parameter in (13) has no physical meaning and all defects correspond to isotropic regions or locally melted regions [20]. The 2D parametrization (14) is a special case of (10) with $q_1 = S \cos 2\phi$, $q_2 = S \sin 2\phi$, $q_3 = 0$. The representation (10) contains more information than (14) since it allows us to investigate structural variations in both q_3 and β^2 , both of which are outside the scope of the 2D representation (14). In particular, the parametrization (10) can resolve the rich biaxial structure of defect cores.

On the faces, $x = 0$ and $x = R$, we set the preferred \mathbf{Q} -tensor to be

$$\mathbf{Q}_s^{(x)} \equiv \mathbf{Q}(0, y, z) = \mathbf{Q}(R, y, z) = \frac{S_{eq}}{3} (-\vec{e}_x \otimes \vec{e}_x + 2\vec{e}_y \otimes \vec{e}_y - \vec{e}_z \otimes \vec{e}_z)\tag{15}$$

where $S_{eq} = \frac{B + \sqrt{B^2 - 24AC}}{4C}$. On the boundaries $y = 0$ (bottom) and $y = R$ (top), we set

$$\mathbf{Q}_s^{(y)} \equiv \mathbf{Q}(x, 0, z) = \mathbf{Q}(x, R, z) = \frac{S_{eq}}{3} (2\vec{e}_x \otimes \vec{e}_x - \vec{e}_y \otimes \vec{e}_y - \vec{e}_z \otimes \vec{e}_z).\tag{16}$$

For systems with Dirichlet conditions, we set $\mathbf{Q} = \mathbf{Q}_s^{(x)}$ and $\mathbf{Q} = \mathbf{Q}_s^{(y)}$ on the respective lateral surfaces. For systems with finite anchoring, we use the prescribed \mathbf{Q} -tensors in (15) and (16) as the preferred state, $\mathbf{Q}_s^{(i)}$, in the

surface anchoring energy density in (8). On the surfaces $z = 0$ and $z = h$, we impose natural boundary conditions i.e. require that

$$\frac{\partial \mathbf{Q}}{\partial \nu} = 0 \quad \text{on } z = 0, h, \quad (17)$$

where $\nu = \pm \vec{e}_z$ is the outward normal to these faces in the (x, y) -plane. This is consistent with the assumed invariance in the z -direction and when combined with planar initial conditions for numerical simulations, ensures that tangent boundary conditions are preferentially imposed on these surfaces.³

3.3 Scaling

Let t be a dimensionless temperature defined to be $t = (T - T_*)/(T_{**} - T_*)$. Therefore, $t(T = T_*) = 0$, $t(T = T_{IN}) = 8/9$ and $t(T = T_{**}) = 1$. We define

$$\tau = 1 + \sqrt{1 - t}. \quad (18)$$

The equilibrium uniaxial order parameter can then be expressed as

$$S_{eq} = S_{**}\tau \quad (19)$$

where $S_{**} = S_{eq}(T_{**}) = \frac{B}{4C}$.

The structural characteristics of static LC equilibria are dictated by a complex interplay between material properties and surface anchoring effects. The key material length-scale is the biaxial correlation length defined to be

$$\xi_b = \frac{\xi_b^{(0)}}{\sqrt{\tau}}, \quad (20)$$

where $\xi_b^{(0)} = 2\sqrt{LC}/B$ is the bare biaxial correlation length [1, 17, 19]. In what follows, we vary temperature and use $\xi_b^{(0)}$ as a characteristic temperature-independent material length scale and for conventional nematics, $\xi_b^{(0)}$ is of the order of a few tens of nm [1, 10]. The strength of the surface interactions is traditionally described in terms of the surface extrapolation length [17, 19].

$$d_e^{(i)} = \frac{L}{w^{(i)}}. \quad (21)$$

Typical values of L are around $10^{-12}N$ and $w^{(i)} \in (10^{-8}, 10^{-3}) N/m$ [23].

We further define the following dimensionless quantities: $\tilde{\mathbf{Q}} = \mathbf{Q}/S_{**}$, $\tilde{x} = x/R$, $\tilde{y} = y/R$, $\tilde{z} = z/R$, $\tilde{\nabla} = R\nabla$, $\tilde{F} = F/F_0$, where $F_0 = LS_{**}^2 R^3/(2\xi_b^{(0)2})$. In what follows, we drop the tildes from the dimensionless variables and all

³We could also use weak anchoring or surface energies on $z = 0$ and $z = h$ to obtain similar results.

subsequent text is to be interpreted in terms of the dimensionless variables. The dimensionless free energy F is given as in (5), and the corresponding dimensionless free energy densities are defined to be

$$\begin{aligned}
f_c &= \frac{\tau}{6} Tr \mathbf{Q}^2 - \frac{2}{3} Tr \mathbf{Q}^3 + \frac{1}{8} (Tr \mathbf{Q}^2)^2 = \\
&\quad \frac{\tau}{3} (q_1^2 + q_2^2 + 3q_3^2) - 4q_3 (q_1^2 + q_2^2 - q_3^2) + \frac{1}{2} (q_1^2 + q_2^2 + 3q_3^2)^2, \\
f_e &= \left(\frac{\xi_b^{(0)}}{R} \right)^2 |\nabla \mathbf{Q}|^2 = 2 \left(\frac{\xi_b^{(0)}}{R} \right)^2 (3|\nabla q_3|^2 + |\nabla q_1|^2 + |\nabla q_2|^2), \\
f_s^{(i)} &= \frac{\xi_b^{(0)2}}{d_e^{(i)} R} Tr (\mathbf{Q} - \mathbf{Q}_s(\varphi_s))^2 = \\
&\quad \frac{2}{3} \left(\frac{\xi_b^{(0)}}{R} \right) \left(\frac{\xi_b^{(0)}}{d_e^{(i)}} \right) (9q_3^2 + 3q_1^2 + 3q_2^2 - 3q_3\tau + \tau^2 - 3q_1\tau \cos(2\varphi_s) - 3q_2\tau \sin(2\varphi_s)).
\end{aligned}$$

In the expression for $f_s^{(i)}$ above, the angle φ_s is either $\varphi_s = 0$ (plates at $y = 0$ and $y = R$) or $\varphi_s = \pi/2$ (plates at $x = 0$ and $x = R$). The strong anchoring limit, $w^{(i)} \rightarrow \infty$, corresponds to vanishing surface extrapolation length, $d_e^{(i)} \rightarrow 0$. We now offer some heuristic insight on pattern formation within the LC wells. On the one hand, if τ is large compared to the scaled elastic constant, $\left(\frac{\xi_b^{(0)}}{R} \right)^2$, then the solution will predominantly minimize the condensation energy i.e. be largely uniaxial with constant order parameter, $S = S_{eq}(T)$ (at least away from defects). On the other hand, if τ and $\left(\frac{\xi_b^{(0)}}{R} \right)^2$ are of comparable magnitude, then elastic distortions and deviations from the condensation energy minima (uniaxial phases with $S = S_{eq}(T)$) e.g. biaxiality, have comparable energetic costs and hence, OR patterns are energetically viable. Similarly, if $\frac{\xi_b^{(0)}}{R} \gg \frac{\xi_b^{(0)}}{d_e^{(i)}}$, then the uniaxial boundary conditions are relatively weakly implemented on the lateral surfaces and the system has greater freedom to adopt almost “spatially homogeneous” or uniform states. This, in turn, allows the system to avoid complex structural transitions, such as the OR phenomenon, for sufficiently weak anchoring. In the next section, we compute phase diagrams for uniaxial-biaxial structural transitions, as a function of R , anchoring strength and temperature, and these phase diagrams corroborate our heuristic insights.

3.4 Euler Lagrange equations

We use standard methods in calculus of variations to compute the Euler-Lagrange equations for extremal points of the LdG energy functional [26]. The Euler-Lagrange equations are a coupled system of elliptic partial differential equations for (q_1, q_2, q_3) as shown below:

$$\left(\frac{\xi_b^{(0)}}{R}\right)^2 \Delta_{\perp} q_1 - \frac{\tau}{6} q_1 + 2q_3 q_1 - \frac{q_1}{2}(3q_3^2 + q_1^2 + q_2^2) = 0, \quad (22)$$

$$\left(\frac{\xi_b^{(0)}}{R}\right)^2 \Delta_{\perp} q_2 - \frac{\tau}{6} q_2 + 2q_2 q_3 - \frac{q_2}{2}(3q_3^2 + q_1^2 + q_2^2) = 0, \quad (23)$$

$$\left(\frac{\xi_b^{(0)}}{R}\right)^2 \Delta_{\perp} q_3 - \frac{\tau}{6} q_3 + \frac{1}{3}(q_1^2 + q_2^2 - 3q_3^2) - \frac{q_3}{2}(3q_3^2 + q_1^2 + q_2^2) = 0, \quad (24)$$

where $\Delta_{\perp} = \frac{\partial^2}{\partial x^2} + \frac{\partial^2}{\partial y^2}$.

The boundary conditions on the plates $x = 0$ and $x = R$ are

$$\frac{\partial q_1}{\partial x} = \mp \frac{R}{d_e^{(x)}} \left(q_1 - \frac{\tau}{2} \right), \quad (25)$$

$$\frac{\partial q_2}{\partial x} = \mp \frac{R}{d_e^{(x)}} q_2, \quad (26)$$

$$\frac{\partial q_3}{\partial x} = \mp \frac{R}{d_e^{(x)}} \left(q_3 - \frac{\tau}{6} \right), \quad (27)$$

$-(+)$ in \mp refers to the right (left) plate and $d_e^{(x)}$ is a measure of the anchoring strength, $w_s^{(x)}$ on each plate.

Similarly, we have the following boundary conditions on the plates $y = 0$ and $y = R$

$$\frac{\partial q_1}{\partial y} = \mp \frac{R}{d_e^{(y)}} \left(q_1 + \frac{\tau}{2} \right), \quad (28)$$

$$\frac{\partial q_2}{\partial y} = \mp \frac{R}{d_e^{(y)}} q_2, \quad (29)$$

$$\frac{\partial q_3}{\partial y} = \mp \frac{R}{d_e^{(y)}} \left(q_3 - \frac{\tau}{6} \right), \quad (30)$$

where $-(+)$ in \mp refers to the top (bottom) plate and $d_e^{(y)}$ is a measure of the relative anchoring strength. The corresponding strong anchoring conditions are $\{q_1 = -\frac{\tau}{2}, q_2 = 0, q_3 = \frac{\tau}{6}\}$ on $y = \{0, R\}$ and $\{q_1 = \frac{\tau}{2}, q_2 = 0, q_3 = \frac{\tau}{6}\}$ on $x = \{0, R\}$. The strong anchoring or Dirichlet conditions can be recovered from (25)-(27) and (28)-(30) in the limit, $d_e^{(x)} \rightarrow 0$ and $d_e^{(y)} \rightarrow 0$ respectively.

We solve the Euler-Lagrange equations (22 - 24) and the boundary constraints (25 - 30) using relaxation methods that have been successful in the study of static LC textures with topological defects [26, 16]. These methods compute the static solutions, (q_1, q_2, q_3) , by mimicking a dynamic gradient-flow like procedure along which the total free energy continuously decreases

till the equilibrium is attained, for an explicitly prescribed initial condition. We use three different kinds of initial conditions: (i) bulk uniaxial alignment along \vec{e}_x , (ii) bulk uniaxial alignment along $\vec{e}_x + \vec{e}_y$, or (iii) the isotropic phase with $\mathbf{Q} = 0$. The \mathbf{Q} -tensor is recovered from the solution, (q_1, q_2, q_3) , by using the parametrization (10). The solutions are robust with respect to different choices of initial conditions and we thus, deduce that they are numerically stable.

4 Results

Macroscopic wells with $R \gg \xi_b^{(0)}$, e.g. micron-sized wells, have already been modelled in detail in [20, 29]. We focus on wells with R comparable to $\xi_b^{(0)}$, in this paper. We use fixed values of L, B, C and hence, the bare biaxial correlation length, $\xi_b^{(0)} = \sqrt{\frac{4LC}{B^2}}$ is a constant throughout this paper. We are interested in structural changes induced by decreasing the ratio $\eta = R/\xi_b^{(0)}$. In simulations, we work with temperatures below T_* , corresponding to $\tau > 2$ (see Eq.(18)). Previous work indicates that biaxial textures are more likely in this low-temperature regime, compared to the high-temperature regime with temperatures above the nematic-isotropic transition temperature. [5, 27]. For example, we carry out illustrative simulations with $\tau = 4$. The same temperature was chosen in the reference [5], where the authors study OR patterns in “classical” hybrid planar cells.

4.1 Diagonal Structure with Defects (DSD)

In Figures 1, 2a and 3, we consider the specific example of a LC well with $R = 4.5\xi_b^{(0)}$, at temperature $\tau = 4$ (with $t = -8$ in (18)), with strong anchoring conditions on the lateral surfaces. In dimensional terms, this would correspond to a well with $R \sim 120 - 150nm$ [10]. The Dirichlet conditions (see (15) and (16)) induce an alignment mismatch along the four vertical edges and consequently, we obtain four line defects along the four vertical edges in the z -direction. In Figure 1 and Figure 2a, we plot the spatial profile of β^2 and in Figure 3, we plot the leading eigenvector (with the largest positive eigenvalue) of the computed \mathbf{Q} -tensor on the cross-section $z = 0$. This eigenvector evidently has a diagonal profile across the square (x, y) cross-section and is strongly reminiscent of the diagonal solution reported in [29, 20]. For a purely uniaxial solution with two degenerate eigenvalues, this eigenvector is simply the director in (2). From Figure 3, it is clear that the leading eigenvector has four defects at the four square vertices and each defect can be regarded as a quarter of $+1$ or -1 -degree defect i.e. the defects at the top left and bottom right are a quarter of $+1$ -degree radial defects and the defects at the bottom left and top right are a quarter of

−1-degree defects. In other words, the leading eigenvector rotates by 90 degrees between a pair of coincident square edges.

In Figures 1 and 2a, we study the structural characteristics of each defect core in terms of β^2 ; the profile is somewhat similar to the classical OR phenomenon studied in [28] etc. We have two intersecting faces at each vertical edge and these faces enforce strictly uniaxial alignments with mutually perpendicular directors, along \vec{e}_x and \vec{e}_y respectively, and positive scalar order parameter. As is evident from the plots, the LC state mediates between the two uniaxial “boundary” states by means of an intermediate biaxial pear-shaped lobe. This biaxial lobe has a rim of maximal biaxiality ($\beta^2 = 1$) and the interior of the lobe has suppressed biaxiality.

In Figure 4, we look at β^2 along a square diagonal on the surface $z = 0$. Our model assumes invariance in the z -direction and hence, it suffices to look at structural characteristics on $z = 0$. We see that β^2 monotonically increases from the vertex to the rim of the biaxial lobe, where $\beta^2 \approx 1$. Here, β^2 has a local maximum and this local maximum is followed by a local minimum near the center of the diagonal. At the center, we are relatively far away from all four square vertices. Then β^2 increases as we approach the diagonally opposite vertex, attains a local maximum at the rim of the diagonally opposite biaxial lobe and finally decreases to zero as we hit the opposite vertex. The β^2 -profile is characterized by two maxima, one for each square vertex, separated by a local minimum. The biaxial defect lobes are asymmetric in shape, in response to the neighbouring defect cores at the different vertical edges. One could define the characteristic linear defect core size to be the distance from the vertex to the nearest rim of maximal biaxiality, measured along the square diagonal. This length is roughly given by ξ_b , as is expected from previously reported results in the literature [5, 17, 16, 19]. We refer to this diagonal profile, with biaxial defect cores near the vertical edges, as the **Diagonal Structure with Defects (DSD)** in the subsequent text.

4.2 Well Order-Reconstruction Structure (WORS)

In this section, we study structural transitions induced by gradually decreasing the ratio $\eta = R/\xi_b^{(0)}$ at a fixed temperature $\tau = 4$. In Figures 2a - 2d, we demonstrate a sequence of $\beta^2(x, y)$ textures obtained by gradually decreasing the ratio, $\eta = R/\xi_b^{(0)}$, with strong anchoring conditions on the lateral surfaces. As η decreases, the biaxial defect lobes become larger (relative to domain size), overlap and eventually connect to yield a 2D star-shaped rim of maximal biaxiality connecting the four vertices. In particular, in Figure 2d, the biaxial ring separates uniaxial states with negative order parameter along the square diagonals from positively ordered uniaxial states near the

square edges (consistent with the boundary conditions).⁴ In this sense, this is a fully 2D OR pattern connecting the four vertices.

If we follow β^2 along a small circular arc, centered around a square vertex, we expect to see the following sequence of characteristic states: (i) positive uniaxiality (positive scalar order parameter) with $\beta^2 = 0$ on the square edge, (ii) maximal biaxiality with $\beta^2 = 1$, (iii) negative uniaxiality (negative scalar order parameter) with $\beta^2 = 0$, (iv) maximal biaxiality with $\beta^2 = 1$, (v) positive uniaxiality with $\beta^2 = 0$ on co-incident edge. By symmetry, we expect to see state (iii) along the square diagonal. In Figure 4, we monitor the structural characteristics along a square diagonal. In Figure 4a, we plot $\beta^2(\eta)$, measured along the square diagonal, for different values of η . For relatively large η , say $\eta = 4.5$, the β^2 -profile has two distinct maxima. This corresponds to the DSD structure reported above. We refer to the two maxima in the regions $\frac{x}{R} < 0.5$ and $\frac{x}{R} > 0.5$ as the *left peak* and *right peak* respectively. As η decreases, the maxima approach each other and coalesce at $\eta = 3.39$. For smaller values of η , say $\eta = 3.3$, the β^2 -maximum is suppressed in magnitude i.e. is smaller than unity. At $\eta = \eta_c = 3.28 \pm 0.01$, β^2 vanishes along the square diagonal (within numerical resolution) and $\beta^2 = 0$ along the diagonal for all $\eta \leq \eta_c$.

In Figure 4b, we compute $\langle \beta^2 \rangle_d$ and β_m^2 as a function of η , corresponding to the average value and the maximal value of β^2 along the square diagonal, respectively. Additionally, we also plot the position of the left peak, denoted by x_m in the region $x \in [0, R/2]$, and the spatial average of β^2 across the square cross-section, denoted by $\langle \beta^2 \rangle$, as a function of η . From Figure 4b, we see that the global OR profile is established at $\eta = \eta_c = 3.28 \pm 0.01$, for which $\beta_m^2 = \langle \beta^2 \rangle_d = 0$. The quantities, $\beta_m^2 = \langle \beta^2 \rangle_d = 0$, for all $\eta \leq \eta_c$. For $\eta \leq \eta_c$, we have an uniaxial LC state, with negative scalar order parameter, along the square diagonal. The corresponding critical well size, $R_c = \eta_c \xi_b^{(0)}$. The average degree of biaxiality, $\langle \beta^2 \rangle(\eta) \approx 0.4$ for $\eta \leq \eta_c$. Since $\beta_m^2 = 0$ for $\eta \leq \eta_c$, x_m is not defined for $\eta < \eta_c$. In what follows, we refer to the star-shaped biaxial OR pattern in Figure 2d as the **Well Order Reconstruction Pattern (WORS)**. We define the onset of the WORS by the critical value of $\eta = \eta_c$ for which $\beta_m^2 = 0$ along the square diagonal on $z = 0$, such that $\beta_m^2 = 0$ for all $\eta < \eta_c$. Our simulations suggest that the DSD-WORS transition is continuously achieved by decreasing η and hence, this is a second-order transition.

4.3 Phase diagrams

In Figure 6a, we numerically study the dependence of the critical well size, $R_c = \eta_c \xi_b^{(0)}$, on the scaled temperature $t = (T - T_*)/(T_{**} - T_*)$, with strong

⁴We interpret regions with $\beta^2 < 0.01$ as being uniaxial for practical purposes. We note that β^2 may not, strictly speaking, vanish at any point inside the well.

anchoring conditions on the lateral surfaces. As t increases towards $T = T^*$ (the temperature at which the isotropic phase loses stability), R_c increases steeply. This steep ascent can be explained on the grounds that the condensation energy penalty for deviations from the uniaxial state with $S = S_{eq}(T)$, decreases as t increases. Hence, biaxial order reconstruction patterns (such as the WORS pattern) become increasingly energetically viable, and hence, observable as t increases.

In Figure 6b, we impose equal anchoring strength on all four lateral surfaces and plot $R_c/\xi_b^{(0)}$ versus the anchoring strength, $\frac{\xi_b^{(0)}}{d_e}$, where $d_e = d_e^{(x)} = d_e^{(y)}$. We work with two different temperatures, $t = -8$ and $t = 0$ respectively. These temperatures correspond to qualitatively different temperature regimes (i.e. $T \ll T_*$ and $T = T_* < T_{IN}$). We observe a slowly increasing profile of R_c versus $\frac{1}{d_e}$ for both temperatures. From the reasoning in Figure 6a, the critical threshold R_c is always higher for higher temperatures. Further, R_c rapidly saturates to a limiting value as $\frac{1}{d_e}$ increases; this limiting value is simply the strong anchoring limit. For $t = -8$ or equivalently for $\tau = 4$, this limiting value is $R_c = 3.28\xi_b^{(0)}$ which coincides with the critical value yielded by the strong anchoring experiments in Figure 4. As $\frac{1}{d_e}$ decreases or equivalently as the anchoring strength decreases, the uniaxial boundary conditions are relatively weakly implemented. This effectively increases the lateral well size since the system has anchoring freedom on the lateral surfaces and hence, does not need to strictly match the preferred uniaxial states on intersecting edges. The WORS profile ceases to exist for anchoring strengths below a certain critical threshold. For such cases, the anchoring conditions are not strong enough to support geometrically imposed frustration.

4.4 Comparison with classical OR structure

In Figures 5a-5c, we compare the WORS pattern with the classical OR pattern reported in a batch of numerical papers [5, 19, 25]. In Figure 5a, we plot $\beta^2(x, y)$ across the square cross-section and observe a distinct star-shaped rim of maximal biaxiality with $\beta^2(x, y) = 1$. In Figure 5b, we plot $-\beta^2(x, y)$ to emphasize the cross-shaped uniaxial ordering (i.e. $\beta^2 = 0$) along diagonals. In Figure 5b, the local eigenframe of \mathbf{Q} coincides with the $\{\vec{e}_x, \vec{e}_y, \vec{e}_z\}$ -laboratory frame and $q_2 = 0$ throughout the square domain in (10).

We can reproduce the classical OR phenomenon in our framework by enforcing conflicting planar and homeotropic anchoring on the plates at $y = 0$ and $y = R_y$ respectively, whilst imposing natural boundary conditions on $x = 0$ and $x = R_x$. Equivalently, we require that

$$\mathbf{Q}(x, 0, z) = \frac{S_{eq}}{3} (2\vec{e}_x \otimes \vec{e}_x - \vec{e}_y \otimes \vec{e}_y - \vec{e}_z \otimes \vec{e}_z) \quad (31)$$

and

$$\mathbf{Q}(x, R_y, z) = \frac{S_{eq}}{3} (2\vec{e}_y \otimes \vec{e}_y - \vec{e}_x \otimes \vec{e}_x - \vec{e}_z \otimes \vec{e}_z). \quad (32)$$

We then progressively decrease the distance, R_y , whilst keeping R_x fixed, at constant temperature $\tau = 4$. At a critical value, $\eta^{(CL)} = 2.45$ (the superscript (CL) marks the *classical* order reconstruction transition), we observe a structural transition into the classical OR pattern and the classical OR pattern is observed for all $\eta < \eta^{(CL)}$. We plot β^2 for the classical OR case in Figure 5c and see two sheets of maximal biaxiality ($\beta^2 = 1$) enclosing a sheet of negative uniaxiality ($\beta^2 = 0$). On comparing Figures 5a and 5c, it is clear that the WORS is qualitatively different to the classical OR pattern; the WORS arises from mutually perpendicular uniaxial alignments on four pairs of coincident edges whereas the classical OR phenomenon arises from mutually perpendicular uniaxial alignments on pairs of parallel edges. Whilst the WORS is observed for $\eta < 3.28$, the classical OR pattern is observed for $\eta < 2.45$ i.e. the WORS pattern has a broader window of stability compared to the classical OR case. We do not have rigorous explanations for these effects but elevated stability thresholds can be potentially attributed to the topological defects present in the WORS. Similar effects have been observed in hybrid cylindrical cells with a boojums on bounding plates [19].

4.5 Interior Defects and the WORS pattern

We demonstrate a DSD-WORS structural transition induced by the introduction of locally melted regions within the DSD structure. This could, in practice, be realized by electronic-beam (e-beam) lithography [9, 31]. E-beam lithography techniques can fabricate sub-10 nm cavities / indentations within square or cuboid geometries. Alternatively, one could introduce a nanoparticle (NP) treated to impose localized isotropic melting at the NP-LC interface. We numerically introduce a melted square cavity of lateral size, r_m , at the center of the well. This requires $\mathbf{Q}(\frac{R}{2} \pm r_m/2, \frac{R}{2} \pm r_m/2, z) = 0$. We take $\tau = 4$, $\frac{R}{\xi_b^{(0)}} = 4.5$, with strong anchoring conditions as in (15) and (16) and no external fields. In dimensional terms, this describes a well with lateral dimensions between 120 – 150 nm [10]. In Figures 7a, 7b and 7c, we numerically compute LC equilibria in the absence and presence of locally melted regions respectively. We recover the familiar DSD in Figure 7a. However, in Fig. 7b, we observe a star-shaped rim of maximal biaxiality, characteristic of the WORS. In Figure 7c, we displace the melted region from the centre towards the lower plate at $y = 0$ and lose the WORS. This suggests that the WORS can be efficiently stabilized if the locally melted region is compatible with the structural symmetry of the WORS.

In Figure 8 we study the effect of the size of the melted region, r_m , on the DSD-WORS structural transition. We follow $\langle \beta^2 \rangle$ as a function of η , for different values of r_m . The curves are almost stationary till they hit

the critical value of $\eta = \eta_c^{(m)}$ and the WORS is stable for all $\eta \leq \eta_c^{(m)}$. It is evident that interior melted regions substantially increase the value of $\eta_c^{(m)}$. For example, we obtain $\eta_c = 3.28 \pm 0.01$ for $r_m = 0$ (no melted region) and $\eta_c^{(m)} = 4.67 \pm 0.01$ for $r_m/R = 0.1$. Hence, the threshold value is increased by roughly 42%.

5 Conclusions

We model and numerically analyze LC textures inside shallow square wells, within the LdG theory for nematic LCs. These wells were first reported in [29] and modelled in the papers [20, 29]. In [20, 29], the authors study large micron-sized wells and perform a strictly two-dimensional study on a square or a rectangle. Our aim is to quantify the effects of confinement, temperature, boundary conditions on the experimentally observed well textures, with special emphasis on new biaxial textures. Our work purely focuses on static LC textures and we do not study field-induced or dynamic effects.

We study shallow square wells with cross-sectional dimension R , where R is measured in units of the bare biaxial correlation length, $\xi_b^{(0)}$. We adopt a 3D parametrization of the LdG \mathbf{Q} -tensor (as opposed to 2D parametrization used in previous work) and assume that all variables only depend on the spatial coordinates, (x, y) , and are independent of the z -coordinate. The boundary conditions on the lateral surfaces, $x = \{0, R\}$ and $y = \{0, R\}$, are imposed either via Dirichlet conditions (as in (15) and (16)) or via a surface anchoring energy as in (8). The Dirichlet conditions in (15) and (16) create line defects along the vertical edges. For wells with $R \gg \xi_b^{(0)}$, the static structures are effectively uniaxial away from the edges and are well described by the *diagonal* and *rotated* solutions in [20, 29]. We study wells with $R = O(\xi_b^{(0)})$ in this paper and obtain two distinct LC states: the DSD and WORS patterns. The DSD pattern is qualitatively similar to the diagonal solution reported in [20, 29] and is only obtained for relatively large wells with R greater than a temperature-dependent critical length, denoted by R_c . The DSD is largely uniaxial away from the vertical line defects, the principal eigenvector has a diagonal profile on the bottom surface ($z = 0$) and there are localized pear-shaped biaxial neighbourhoods near each vertical edge, the rims of which exhibit maximal biaxiality. We believe that the macroscopic properties of the DSD solution are qualitatively similar to the diagonal solution in [20, 29] and our 3D modelling recovers the biaxial structure of the defect cores, which is expected but not reported in previous work.

We progressively decrease the ratio $\eta = \frac{R}{\xi_b^{(0)}}$ and at a critical value $\eta = \eta_c(\tau)$, (e.g. $\eta_c(4) = 3.28 \pm 0.01$) we observe a global WORS pattern with a star-shaped ring of maximal biaxiality connecting the four vertices of the

square cross-section on $z = 0$. This ring is repeated throughout the height of the cell. The biaxial ring separates uniaxial states with negative order parameter along the square diagonals from uniaxial states with positive order parameter outside the ring respectively. The critical value depends on the temperature and on the anchoring strength on the lateral surfaces. We define a quantitative criterion for the onset of the WORS pattern, in terms of the maximum value of β^2 (β_m^2) along a square diagonal. The onset is defined by the critical value of η , denoted by η_c , for which $\beta_m^2 = 0$ and the WORS is the only observed state for $\eta \leq \eta_c$. The DSD-WORS transition is a continuous transition and we interpret stable states with $\beta_m^2 = \langle \beta^2 \rangle_d \neq 0$ for $\eta > \eta_c$ as being a variant of the DSD state. Further, we study the dependence of η_c on the temperature and the anchoring strength on the lateral surface, with no external fields in Figure 6. Numerical simulations show that interior locally melted regions, do not substantially alter the symmetry of WORS but substantially increase the stability window of the WORS.

Next, we discuss the reliability of the mesoscopic LdG theory for sub-micron and nano-scale systems, such as the model systems studied here. Firstly, the LdG theory has been used, with success, to study nano-scale defect structures [28] and the OR phenomenon in nano-confined systems [5, 25, 16, 17, 18]. Recent experimental work in [13, 15, 18] confirms that mesoscopic modelling yields surprisingly good predictions, even on the nm scale. For example, a simplified LdG-type model can reproduce experimental observations in cylindrical nano-channels with radius of order $\sim 5nm$, in [13]. Secondly, our model predicts that WORS patterns are observable for $R \sim 3.28\xi_b^{(0)}$ for temperature, $\tau = 4$. As stated in the Introduction, $\xi_b^{(0)}$ is typically tens of nm [6]. Recent experimental work in a π -cell geometry, with conventional optical microscopy techniques, suggests that $\xi_b^{(0)} \sim 33 \pm 9 nm$ [10]. This would correspond to cell sizes, $R \sim 78 - 138 nm$, and such dimensions are widely modelled by LdG approaches in the community. At higher temperatures, the WORS is observable in even larger wells, with dimensions $R \sim 240 - 420 nm$ (see Figure 6a). Furthermore, polymeric LCs can have correlation lengths which are an order of magnitude larger than that of their conventional nematic counterpart [7]. In such cases, the WORS pattern would be observed even in micron-scale wells.

Biaxial patterns have been extensively found in the literature; see [16, 17, 19, 24, 25, 28] etc., the reported textures are heavily localized near defects or boundaries. The WORS is a new global 2D biaxial texture predicted for sub-micron and nano-systems with moderate to strong anchoring conditions, for temperatures below the supercooling temperature. The WORS is qualitatively different to the reported diagonal and rotated solutions in [29, 20]. As such, we expect that it will offer different optical properties and new responses to electric fields etc. Our numerical work can give engineers quantitative information about the scale and shape of prototype devices

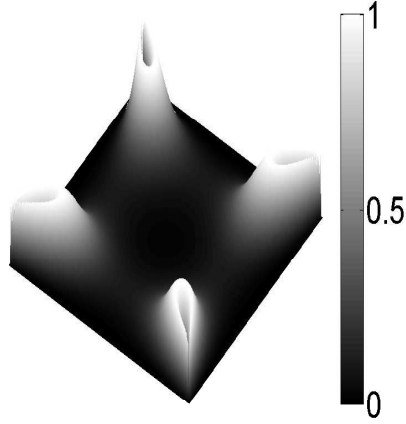


Figure 1: The degree of biaxiality, $\beta^2(x, y)$, in a square well with $R/\xi_b^{(0)} = 4.5$, $\tau = 4$ and strong anchoring conditions. The shading code for $\beta^2 \in [0, 1]$ is on the right side.

that can support the WORS pattern. Once such devices are actually engineered, one could experimentally measure the electro-optical responses of the WORS pattern, possibly by analogy with similar experimental work on OR patterns in one-dimensional hybrid cells in [3, 12]. Whilst our work sets up a sound foundation for the study of the WORS pattern, there are several directions for further study which include (i) field-induced effects, (ii) dynamic phenomena, (iii) effects of elastic anisotropy, (iv) non-cuboid geometries and (v) inclusion of interior defects. A comprehensive study on these lines can yield vital information about systems with intrinsic biaxiality and how biaxiality can be exploited for new nano-scale applications.

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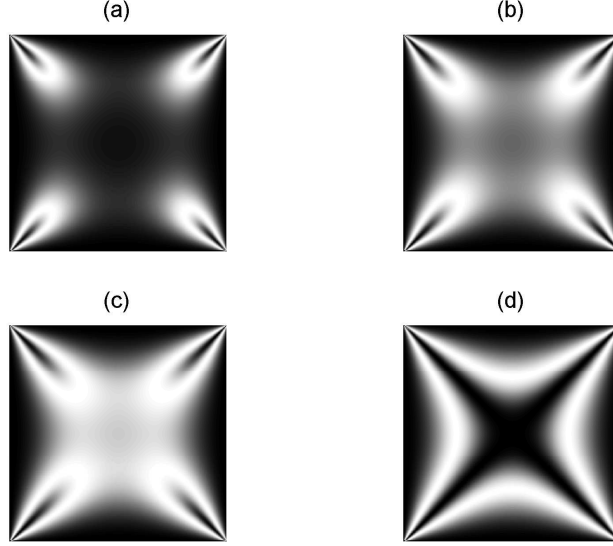


Figure 2: Plots of $\beta^2(x, y)$ with decreasing $\eta = R/\xi_b^{(0)}$, at a fixed temperature $\tau = 4$ and strong anchoring conditions. (a) $\eta = 4.5$, (b) $\eta = 3.8$, (c) $\eta = 3.6$, (d) $\eta = 3$. We observe the WORS below the critical value $\eta_c = 3.28 \pm 0.01$.

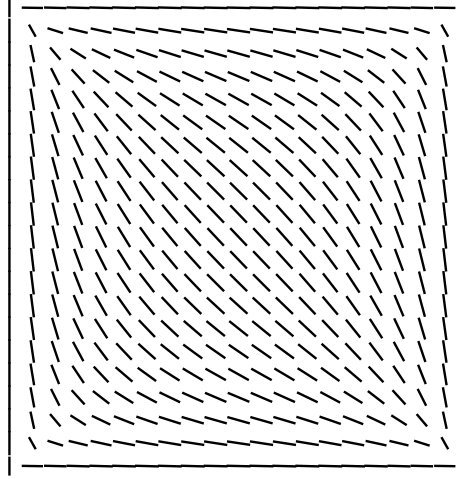


Figure 3: The maximal eigenvector (eigenvector with the largest positive eigenvalue) of the \mathbf{Q} -solution with $R/\xi_b^{(0)} = 4.5$, $\tau = 4$ and strong anchoring conditions. In the uniaxial limit, this eigenvector is simply the nematic director field. The corresponding $\beta^2(x, y)$ dependence is plotted in Fig. 1.

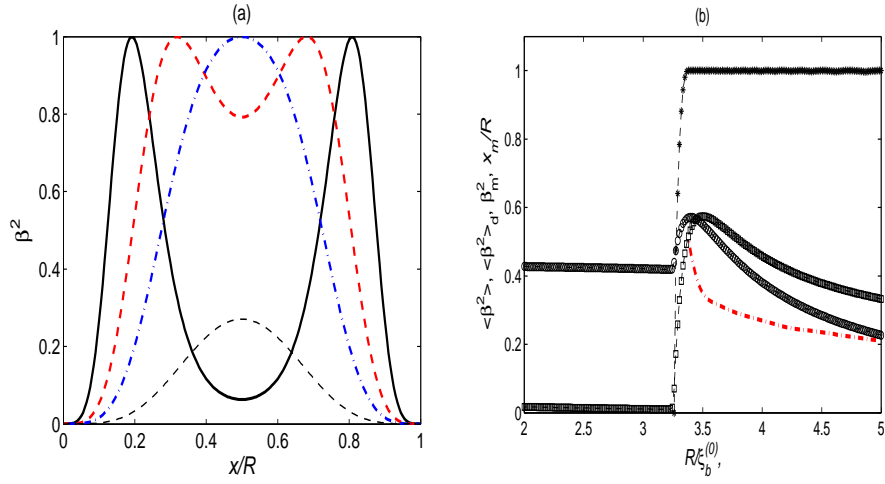


Figure 4: Onset of WORS structure with $\tau = 4$ and strong anchoring conditions. (a) Plots of $\beta^2(x)$ along the well diagonal, where x/R is the scaled distance along the square diagonal measured from the bottom left vertex. $\eta = 4.5$, thick full line; $\eta = 3.5$, thick dashed line; $\eta = 3.39$, thick dash-dash line; $\eta = 3.3$, thin dashed line with maximum at $\beta^2 < 0.5$. (b) Structural characteristics of WORS: $\langle \beta^2 \rangle_d$ (open squares), $\langle \beta^2 \rangle$ (open spheres), β_m^2 (asterisks) and x_m (dashed line) as a function of η . The WORS is defined by $R = R_c$ for which $\beta_m^2 = \langle \beta^2 \rangle_d = 0$, $\beta_m^2 = 0$ for $R < R_c$ and x_m is not defined for $\eta < \eta_c$.

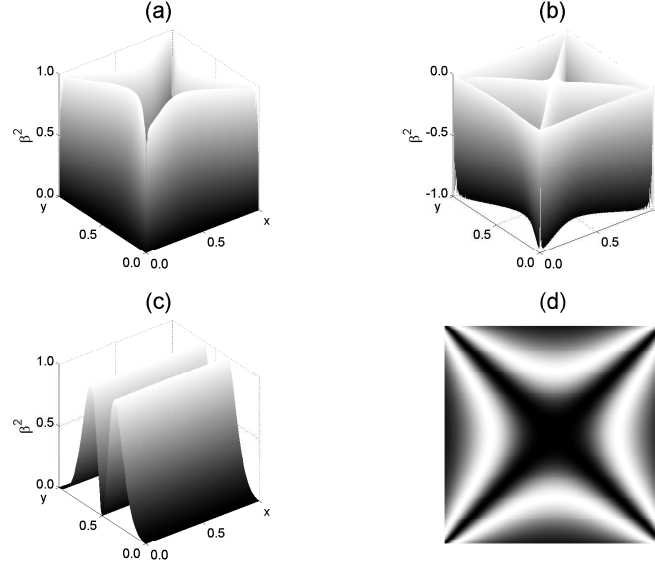


Figure 5: $\beta^2(x, y)$ plots for OR patterns with $\tau = 4$. (a) and (b): the WORS texture in a square well; $\eta = 3$, strong anchoring condition. (c) the *classical* OR structure within a hybrid cell; $\eta = 2.5$, strong anchoring condition. (d) WORS (top view with darker regions corresponding to suppressed biaxiality), $\eta = 3$ with weak anchoring condition ($R/d_e = 5$).

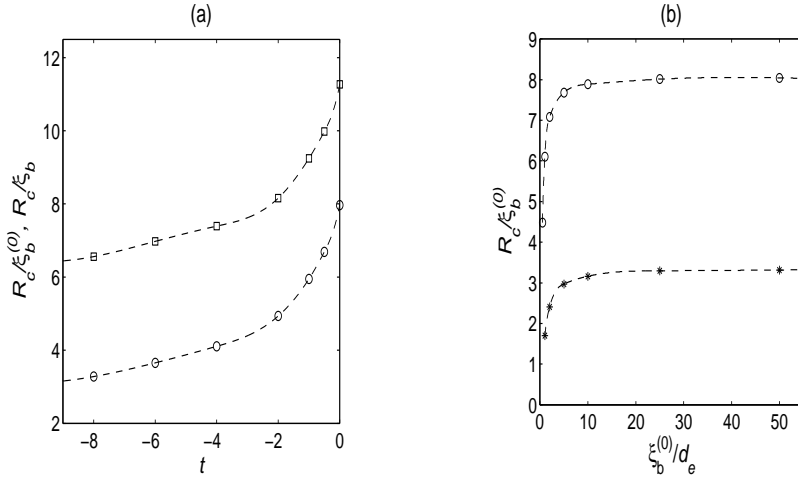


Figure 6: Effect of temperature and anchoring strength on the stability of WORS. Symbols designate points where thresholds are calculated. Lines are guides for the eye. (a) Temperature variation of $\eta_c^* = \sqrt{\tau}\eta_c = R_c/\xi_b$ (squares) and $\eta_c = R_c/\xi_b^{(0)}$ (spheres) with strong anchoring condition. (b) Impact of anchoring strength on η_c for $t = -8$ (asterisk symbols) and $t = 0$ (spheres).

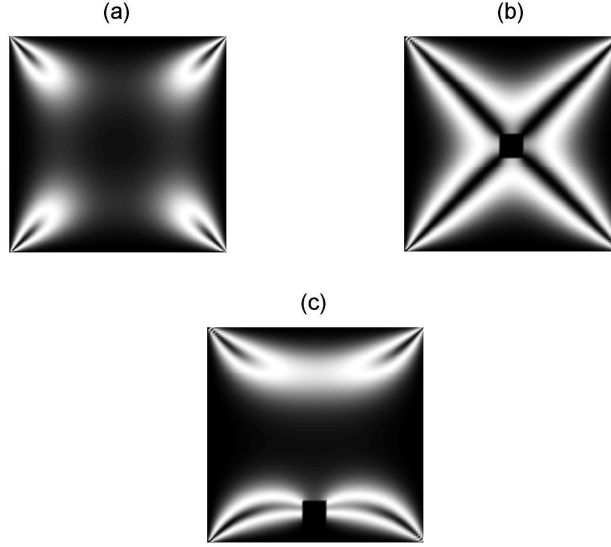


Figure 7: Effect of interior melted region on stability of nematic structures with parameters $R/\xi_b^{(0)} = 4.5$, $\tau = 4$, strong anchoring. (a) The DSD in the absence of a melted region. (b) The WORS with a melted region at the center of the well. (c) The melted region is displaced below the center of the well.

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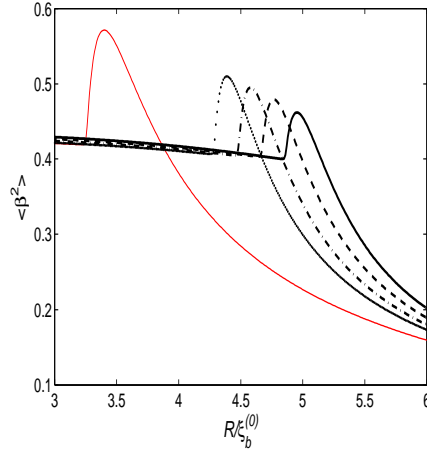


Figure 8: The average degree of biaxiality as a function of $\eta = R/\xi_b^{(0)}$ for structures (i) without (full thin line with maximum at $\frac{R}{\xi_b^{(0)}} \approx 3.5$) and with melted region (thick lines) revealing the DSD-WORS structural transition. The plots are computed with $\tau = 4$ and strong anchoring conditions. The linear size r_m of square melted region is equal to (i) $r_m = 0$, (ii) $r_m/R = 0.1$ (thick full line), (iii) $r_m/R = 0.08$ (thick dashed line), (iv) $r_m/R = 0.06$ (thick dash-dash line), (v) $r_m/R = 0.04$ (thick dotted line). The structural transition takes place at (i) $\eta_c = 3.28 \pm 0.01$, (ii) $\eta_c^{(m)} = 4.86 \pm 0.01$, (iii) $\eta_c^{(m)} = 4.67 \pm 0.01$, (iv) $\eta_c^{(m)} = 4.48 \pm 0.01$, (v) $\eta_c^{(m)} = 4.29 \pm 0.01$.

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